

Plasmon–Exciton Coupling Using DNA Templates

Eva-Maria Roller,[†] Christos Argyropoulos,[‡] Alexander Högele,[†] Tim Liedl,[†] and Mauricio Pilo-Pais^{*,†}

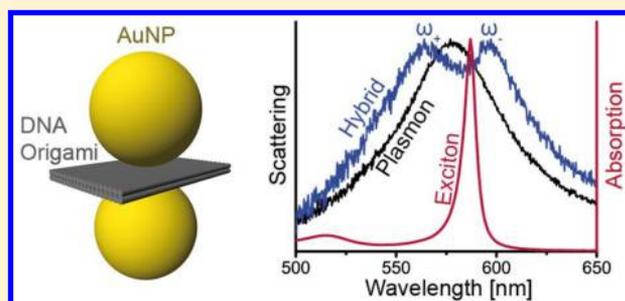
[†]Faculty of Physics and Center for NanoScience (CeNS), Ludwig-Maximilians-Universität (LMU) München, Geschwister-Scholl-Platz 1, 80539 Munich, Germany

[‡]Department of Electrical and Computer Engineering, University of Nebraska-Lincoln, Lincoln, Nebraska 68588, United States

S Supporting Information

ABSTRACT: Coherent energy exchange between plasmons and excitons is a phenomenon that arises in the strong coupling regime resulting in distinct hybrid states. The DNA-origami technique provides an ideal framework to custom-tune plasmon–exciton nanostructures. By employing this well controlled self-assembly process, we realized hybrid states by precisely positioning metallic nanoparticles in a defined spatial arrangement with fixed nanometer-sized interparticle spacing. Varying the nanoparticle diameter between 30 nm and 60 nm while keeping their separation distance constant allowed us to precisely adjust the plasmon resonance of the structure to accurately match the energy frequency of a J-aggregate exciton. With this system we obtained strong plasmon–exciton coupling and studied far-field scattering at the single-structure level. The individual structures displayed normal mode splitting up to 170 meV. The plasmon tunability and the strong field confinement attained with nanodimers on DNA-origami renders an ideal tool to bottom-up assembly plasmon–exciton systems operating at room temperature.

KEYWORDS: DNA origami, plexcitons, excitons, plasmons, J-aggregates, Rabi splitting



Nanoparticles (NPs) subjected to light excitation exhibit collective oscillations of electrons (*plasmons*), which in turn can greatly affect the behavior of quantum emitters positioned in nearby locations. The resulting plasmon–exciton coupling is of interest as it may facilitate studies of fundamental quantum phenomena such as coherent energy exchange, entanglement, and cavity quantum electrodynamics.¹ Potential applications of strongly coupled-exciton systems include artificial light harvesting,² threshold-less lasing, or their use in quantum information processing.³ The degree of interaction between plasmons and quantum emitters can be classified based on their coupling strength (g), displaying different signatures in the far-field scattering spectra, such as enhanced absorption dip, Fano resonance, or Rabi splitting.³ Although these effects are usually associated with quantum-mechanical phenomena, they can be qualitatively described by classical electrodynamics.^{3–5} Plasmon frequencies can be tuned by varying the metallic NP size, geometry, interparticle separation, and their two- or three-dimensional arrangement. Moreover, near-field enhancement can be obtained using small gaps among metallic NPs or using structures with sharp morphology.⁶ If an exciton is placed in regions with enough field confinement, it is possible to achieve the necessary coupling strength to reach the regime of strong coupling, which results in a normal mode splitting, in close analogy to a coupled harmonic oscillator.³ Our focus is on the strong coupling regime where the energy exchange between the plasmon and the exciton results in distinct hybrid modes, the so-called *plexciton* states.

Experimental realizations of plasmon–exciton coupling include work on metallic films,⁷ lithographic constructs,^{8,9} and individual colloids.¹⁰ Even though complex structures can be fabricated using lithographic techniques and have already been used to promote plasmon–exciton coupling,^{8,9} this methodology is limited in the minimum feature size. In addition, metallic structures produced by lithography exhibit greater plasmon damping due to their surface roughness and inherent grain boundaries. All of these aspects lower the quality factor (Q) and the near-field enhancement, decreasing the interaction strength one could potentially achieve with top-down fabricated structures. To circumvent these limitations, one can resort to colloidal NPs, which are routinely synthesized in well-defined sizes and feature less ohmic losses due to their higher crystallinity as compared to top-down structures. Consequently, individual nanocrystals such as gold shells,¹¹ silver rods,¹⁰ and silver triangles^{12–14} have already been reported to display plexcitonic signatures in the presence of J-aggregates. These unmodified colloids, however, lack the ability to assemble into complex plasmonic designs that are required for specific applications.² More importantly, individual colloids do not take advantage of the additional field confinement that results from bringing together two or more closely spaced NPs. In addition, silver colloids are known to

Received: July 20, 2016

Revised: August 16, 2016

Published: August 17, 2016

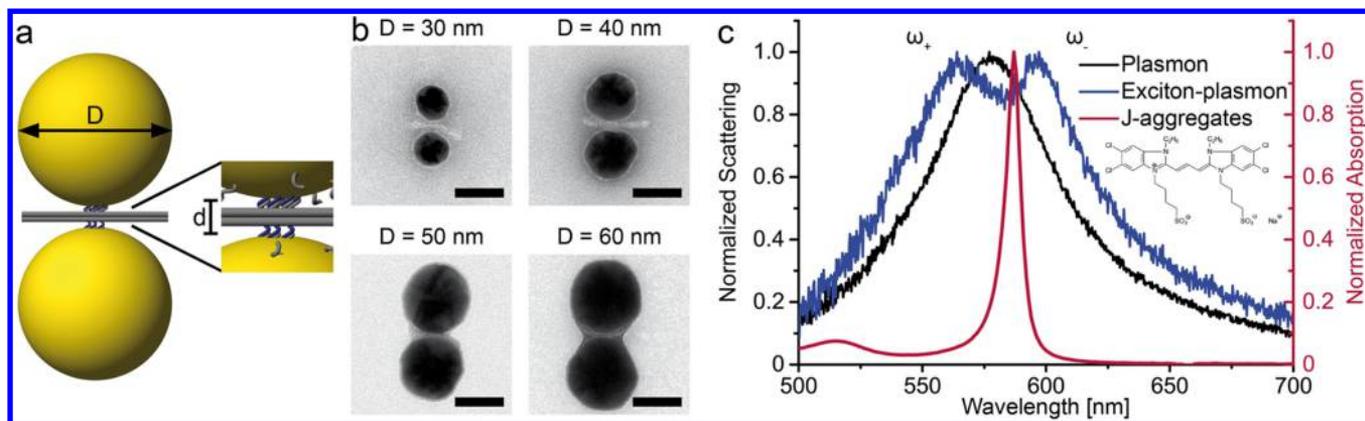


Figure 1. (a) Schematic of a two-layer DNA-origami sheet templating a gold nanoparticle dimer with a designed separation of ~ 5 nm. (b) TEM images of DNA sheets accommodating NPs of different diameters ranging from 30 to 60 nm. Scale bars are 40 nm. (c) Typical spectra of 40 nm dimer with (dark blue) and without (black) J-aggregates. J-aggregate absorption is depicted in red. Inset shows the chemical structure of the molecular exciton used in this work.

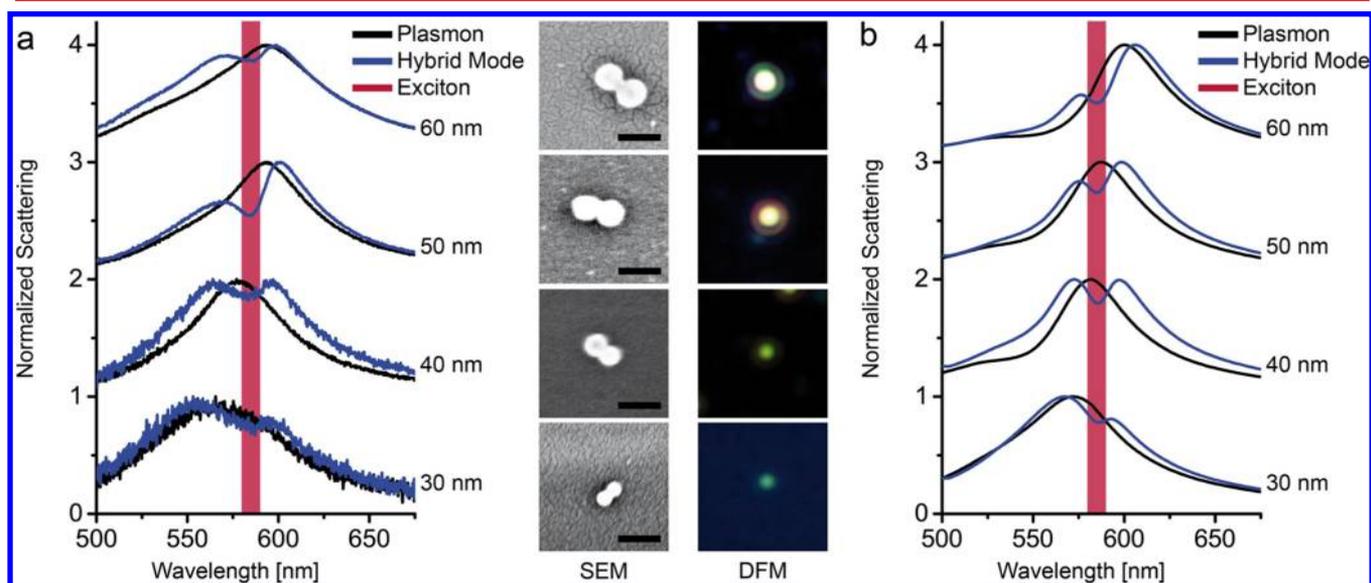


Figure 2. (a) Normalized scattering spectra before (dark-blue) and after (black) photobleaching the J-aggregate for 30, 40, 50, and 60 nm NPs dimers. The spectral region shaded in red covers the energy resonance of the J-aggregate. The right panels show the structures corresponding to each spectrum under SEM and darkfield microscopy. Scale bars: 100 nm. (b) Numerical simulations show excellent agreement with the experimental data.

oxidize, making the use of colloidal gold the preferred choice for plasmon–exciton systems. Very recently, plasmon–exciton coupling using individual molecules in combination with a nanoparticle-on-mirror (NPoM) configuration,¹⁵ as well as coupling between individual colloidal QDs with lithographically produced silver bow-tie antennas¹⁶ have been reported, highlighting plasmonic cavities as promoters of strong light-matter interactions.

DNA-origami is a technique routinely used to fabricate structures with nanoscale dimensions (~ 100 nm) and programmable designs.^{17,18} In a one-pot reaction, a long viral single-stranded DNA (ssDNA) scaffold (~ 7 k bases) is folded by the help of ~ 200 complementary short synthetic ssDNA oligonucleotides. These structures can be used as templates with sequence-specific DNA binding sites, where nano-components functionalized with complementary DNA sequences can be attached to the binding sites (Figure 1a).^{19,20} DNA-templated metallic structures have already been tailored to affect the optical properties of nearby components such as custom-tuned “hot spots” for surface-enhanced Raman

scattering (SERS),^{21–24} enhancement and quenching of fluorophores,^{25,26} and colloidal quantum dots.^{27,28} In addition, the DNA-origami technique has been successfully used to tailor light, displaying strong circular dichroism²⁹ as well as magnetic resonances.³⁰

Here, we demonstrate strong coupling between plasmons and excitons (J-aggregates) at room temperature and optical frequencies by exploiting the position accuracy that is achievable with the DNA-origami technique. This technique provides unprecedented control in the design of plexcitonic systems, bringing this technology one step closer to practical applications as compared to all previously proposed plexcitonic designs. By attaching pairs of colloidal gold nanocrystals to a DNA origami template, we fabricated a nanoantenna configuration with a fixed intergap distance of ~ 5 nm. The resonance frequency of the longitudinal plasmon mode of our constructs scales with the NP size and thus can be tuned across and even matched with the resonance of the desired exciton. This allowed us to observe normal mode splitting in the far-field scattering of individual constructs.

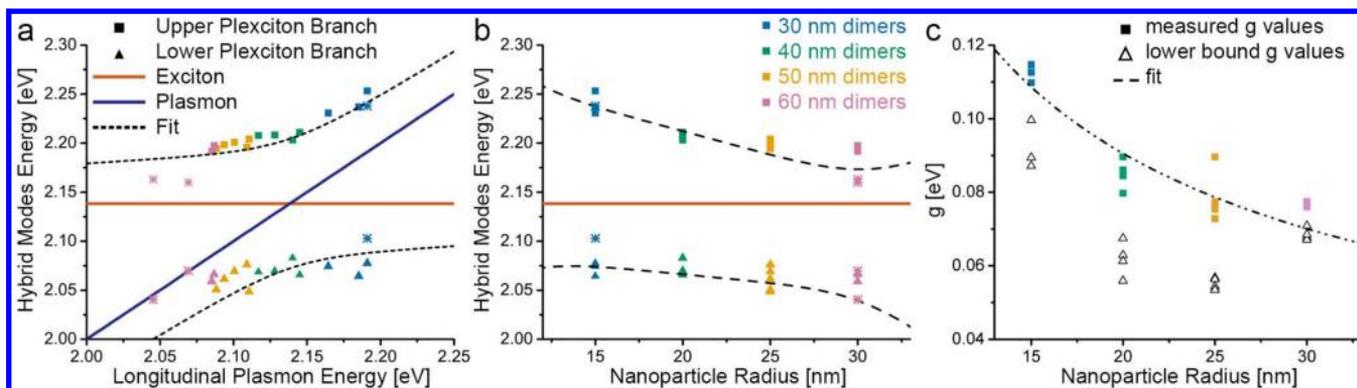


Figure 3. (a) Hybrid plasmon–exciton state energies are plotted as a function of their corresponding plasmon resonance and (b) as a function of the NP radius, showing the anticrossing behavior. Dashed lines are fits of the eigenvalues of a two-coupled harmonic oscillator with complex frequencies. Stars represent structures that do not fulfill the strong coupling condition and were not used in the fitting procedure. Both fittings (a and b) yield a Rabi splitting of $\Omega \sim 150$ meV. (c) Coupling constant (g) values obtained for individual structures reveal a scaling of $g \propto 1/R^n$, $n = 0.63 \pm 0.08$ in very good agreement with the expected $g \propto 1/\sqrt{V_{\text{eff}}} = 1/R^{0.5}$. Dimers composed of smaller NP sizes display a larger coupling constant g . Triangles depict the lower bound for the strong coupling given by $g^2 > (\Gamma_p^2 + \Gamma_{\text{qe}}^2)/16$.

Our DNA-templated nanodimer assemblies were fabricated using pairs of 30, 40, 50, or 60 nm diameter gold NPs functionalized with DNA linkers complementary to specific binding sites on a two-layered DNA-origami sheet (Figure 1a). Transmission electron microscopy (TEM) images of gel-purified structures reveal high yields (Supporting Figures S1–S3) of correctly assembled particle dimers with designed interparticle gap of 5 nm. For this particular gap size we find that dimers built from 40 nm NPs are in closest spectral resonance with the exciton frequency of the cyanine-based dye used in this work (CAS No. 18462-64-1, FEW Chemicals GmbH). This methanol soluble dye readily stacks to form J-aggregates when dissolved in water. When absorbed to glass substrates, we find thin layers of J-aggregates to exhibit a scattering peak at 580 nm (2.14 eV) and a narrow fwhm line width of 30 meV (Supporting Figure S5b). For our measurements of combined plasmon-exciton systems, the assembled structures were deposited on a glass substrate and then immersed in a J-aggregate water bath solution (50 μM). After overnight incubation, the samples were blown with nitrogen, flushing out most of the J-aggregate excess except at the location of the NP dimers. Far-field scattering measurements on individual structures were then performed using a home-built darkfield microscope (Supporting Figure S4 and Note 2). After recording the spectral response of the hybrid structures, the samples were exposed for 1 h to continuous white light illumination under a 100 \times objective to completely photobleach the J-aggregates.¹² This permitted us to additionally record the plasmon resonance of the structures without the contribution of the excitons. Figure 1c shows the far-field scattering spectra of a single AuNP dimer assembled using DNA-origami with J-aggregates before (dark-blue line) and after (black line) photobleaching the excitons.

Spectral red-shifts are more pronounced on dimers with bigger NP sizes due to stronger interparticle coupling. Thus, detuning of the plasmon mode with respect to the exciton resonance was achieved by building dimers with NPs sizes ranging from 30 to 60 nm (Figure 2a), while using the same origami design and thus a constant interparticle gap. This allowed us to tune the plasmon resonance wavelength between 2.05 and 2.20 eV across the exciton resonance at 2.14 eV. As a result of the coupling, the scattering spectrum splits into hybrid

states of lower (ω_-) and higher (ω_+) energies. Polarization-resolved measurements show that only the longitudinal mode couples, as only this mode matches the exciton resonance (Supporting Figure S5b). For each particle size, we measured the scattering spectra of several individual structures and present exemplary spectra for all sizes in Figure 2a. Numerical simulations show excellent agreement with the experimental data (Figure 2b, Supporting Note 4). Using the data collected for Figure 2a, we followed the position evolution of the upper ω_+ and the lower ω_- hybrid states as a function of the longitudinal plasmon mode and NP radius (R). As expected, the energy positions display a pronounced avoided crossing, characteristic of strong coupling (Figure 3).³

The Rabi frequency (Ω) corresponds to the spectral separation of the normal modes ($\Delta\omega$) when the plasmon and the exciton are at perfect resonance. To extract its value, we modeled the system as a two coupled harmonic oscillators with complex frequencies $\tilde{\omega} = \omega + i\Gamma/2$. The resulting complex eigenvalues are³¹

$$\tilde{\omega}_{\pm} = \frac{\tilde{\omega}_p + \tilde{\omega}_{\text{qe}}}{2} \pm \sqrt{g^2 + \frac{(\tilde{\omega}_p - \tilde{\omega}_{\text{qe}})^2}{4}} \quad (1)$$

where $\tilde{\omega}_p$ and $\tilde{\omega}_{\text{qe}}$ are the complex frequencies of the plasmon and quantum emitter (J-aggregate), respectively, and g is the coupling constant. The position of the hybrid modes (ω_{\pm}) as well as the fwhm line-widths of the exciton (Γ_{qe}) and the plasmon (Γ_p) were extracted by Lorentzian fitting of the corresponding scattering spectra, as described in Supporting Note 3. The spectral separation between the upper and lower modes resonances, $\Delta\omega$, is then given by (Supporting Note 3):

$$(\Delta\omega)^2 = \sqrt{(\omega_p - \omega_{\text{qe}})^2 (\Gamma_p - \Gamma_{\text{qe}})^2 + \left(4g^2 + (\omega_p - \omega_{\text{qe}})^2 - \frac{(\Gamma_p - \Gamma_{\text{qe}})^2}{4}\right)^2} \quad (2)$$

eq 2 reduces to the commonly used Rabi splitting $\Omega = 2\sqrt{g^2 - (\Gamma_p - \Gamma_{\text{qe}})^2/16}$ when $\omega_p = \omega_{\text{qe}}$, which sets a threshold of $g^2 > (\Gamma_p - \Gamma_{\text{qe}})^2/16$ to ensure the splitting is real valued. An indication that the strong coupling regime has been reached³ is given by $g^2 > (\Gamma_p^2 + \Gamma_{\text{qe}}^2)/16$, which shows that the

splitting between the new modes is greater than their line width. This sets a lower bound value of $g > 60$ meV for the structures in resonance with the exciton ($\Gamma_{p,r=20\text{nm}} = 240$ meV). To extract the coupling constant g of our system, we first performed a fitting on the upper ω_+ and lower ω_- modes as a function of the plasmon frequency ω_p (Figure 3a). This procedure assumes a constant g for all particle sizes. The extracted exciton–plasmon coupling g_{fit} is ~ 90 meV, which results in a Rabi splitting of $\Omega \sim 150$ meV.

To account for the varying $g(R)$ as the NP radius is changed, we then fitted the exponential function $g = aR^n$ (Figure 3c) using radial-dependent parameters ($\Gamma_{p(R)}, \omega_{p(R)}$) extracted from the recorded data (Supporting Figure S6). Here, a and n are fitting parameters. This analysis revealed a coupling constant which scales with the NP radius as $g \sim 1/R^n$, $n = 0.63 \pm 0.08$, showing that higher coupling constants are obtained for smaller NP sizes. At the expected anticrossing position ($R = 20$ nm), both fitting procedures provide an equal value of $g \sim 90$ meV. Note that in our analyses only those spectra with a real valued coupling constant were taken into account when considering eq 2. Structures with values far from resonance (depicted with asterisks in Figure 3) only exhibited a Fano-like signature or absorbance dip enhancement. Figure 3c displays the g dependence on the radius and clearly shows that dimers with reduced NP size exhibit larger coupling constants. The g values were extracted by replacing the individual R , ω_p , $\Delta\omega$, and Γ_p parameters in eq 2 for each measured NP dimer. Following ref 32, we approximated the effective mode volume of two closely spaced NPs to be a cylinder with a circular base of diameter \sqrt{Rd} (width of the induced surface charge) and height d (gap distance), $V_{\text{eff}} \propto Rd^2$.³² The measured $g \sim 1/R^n$ with $n = 0.63 \pm 0.08$ is in very good agreement with the expected scaling of $g \propto 1/\sqrt{V_{\text{eff}}} \sim 1/R^{0.5}$. In summary, we observe that smaller particles exhibit the strongest coupling. However, the steep decrease in scattering of smaller NP systems makes the study of individual structures below 30 nm in size challenging. Moreover, we also expect that as the NP size is further reduced, surface scattering damping would start to dominate. Thus, there is an ideal NP size where the coupling strength is maximal. This regime was not accessible with our current experimental setup.

We have successfully demonstrated that DNA templates can be used to rationally engineer plexcitonic systems that display hybridized modes between plasmons and molecular excitons (J-aggregates) at room temperature. Our structures are programmed to self-assemble in solution and take full advantage of the field confinement produced by closely spaced metallic colloidal nanocrystals. The coupled plasmon mode can be custom-tuned to be in resonance with the exciton of interest by setting the desired interparticle separation and nanoparticle size. Moreover, one could further exploit the full addressability of the DNA-origami technique to incorporate and precisely position additional nanocomponents, such as individual dyes or quantum dots. As such, the DNA-origami technique provides an unparalleled control in the fabrication of plexcitonic systems and represents a promising platform to achieve fully integrated nanobreadboards and quantum nanocircuits. In future work, we will further investigate the coherent energy exchange between the plasmon and the exciton of our hybrid systems via second-order photon correlation spectroscopy. The design flexibility and the parallel assembly formation of DNA templates are

ideally suited to study plasmon–exciton coupling and to fabricate complex structures for optical applications.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.nanolett.6b03015.

Materials, detailed experimental methods, data analysis, and numerical calculation procedures (PDF)

■ AUTHOR INFORMATION

Corresponding Author

*E-mail: m.pilopais@lmu.de.

Author Contributions

E.M.R., A.H., T.L., and M.P. conceived the experiment. E.M.R. and M.P. conducted the experiments and analyzed the results. C.A. performed the numerical calculations. All authors interpreted the data and reviewed the manuscript.

Notes

The authors declare no competing financial interest.

■ ACKNOWLEDGMENTS

This work was funded by the Volkswagen Foundation, the DFG through the Nanosystems Initiative Munich (NIM) and the ERC through the Starting Grant ORCA. A.H. acknowledges funding by the ERC starting grant No. 336749. C.A. would like to acknowledge support by the Office of Research and Economic Development at University of Nebraska Lincoln and the NSF Nebraska MRSEC.

■ REFERENCES

- (1) Tame, M. S.; McEnery, K. R.; Özdemir, K.; Lee, J.; Maier, S. A.; Kim, M. S. *Nat. Phys.* **2013**, *9*, 329–340.
- (2) Gonzalez-Ballester, C.; Feist, J.; Moreno, E.; Garcia-Vidal, F. J. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2015**, *92*, 121402.
- (3) Törmä, P.; Barnes, W. L. *Rep. Prog. Phys.* **2015**, *78*, 013901.
- (4) Savasta, S.; Saija, R.; Ridolfo, A.; Di Stefano, O.; Denti, P.; Borghese, F. *ACS Nano* **2010**, *4*, 6369–6376.
- (5) Fauchaux, J. A.; Fu, J.; Jain, P. K. *J. Phys. Chem. C* **2014**, *118*, 2710–2717.
- (6) Novotny, L.; van Hulst, N. *Nat. Photonics* **2011**, *5*, 83–90.
- (7) Bellessa, J.; Bonnand, C.; Plenet, J. C.; Mugnier, J. *Phys. Rev. Lett.* **2004**, *93*, 036404.
- (8) Bellessa, J.; Symonds, C.; Vynck, K.; Lemaître, A.; Brioude, A.; Beaur, L.; Plenet, J. C.; Viste, P.; Felbacq, D.; Cambil, E.; Valvin, P. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2009**, *80*, 033303.
- (9) Schlather, A. E.; Large, N.; Urban, A. S.; Nordlander, P.; Halas, N. J. *Nano Lett.* **2013**, *13*, 3281–3286.
- (10) Zengin, G.; Johansson, G.; Johansson, P.; Antosiewicz, T. J.; Käll, M.; Shegai, T. *Sci. Rep.* **2013**, *3*, 3074.
- (11) Fofang, N. T.; Grady, N. K.; Fan, Z.; Govorov, A. O.; Halas, N. J. *Nano Lett.* **2011**, *11*, 1556–1560.
- (12) Zengin, G.; Wersäll, M.; Nilsson, S.; Antosiewicz, T. J.; Käll, M.; Shegai, T. *Phys. Rev. Lett.* **2015**, *114*, 157401.
- (13) Balci, S. *Opt. Lett.* **2013**, *38*, 4498.
- (14) DeLacy, B. G.; Miller, O. D.; Hsu, C. W.; Zander, Z.; Lacey, S.; Yagloski, R.; Fountain, A. W.; Valdes, E.; Anquillan, E.; Soljačić, M.; Johnson, S. G.; Joannopoulos, J. D. *Nano Lett.* **2015**, *15*, 2588–2593.
- (15) Chikkaraddy, R.; de Nijs, B.; Benz, F.; Barrow, S. J.; Scherman, O. A.; Rosta, E.; Demetriadou, A.; Fox, P.; Hess, O.; Baumberg, J. J. *Nature* **2016**, *535*, 127–130.
- (16) Santhosh, K.; Bitton, O.; Chuntunov, L.; Haran, G. *Nat. Commun.* **2016**, *7*, ncomms11823.

- (17) Rothemund, P. W. K. *Nature* **2006**, *440*, 297–302.
- (18) Douglas, S. M.; Marblestone, A. H.; Teerapittayanon, S.; Vazquez, A.; Church, G. M.; Shih, W. M. *Nucleic Acids Res.* **2009**, *37*, 5001–6.
- (19) Schreiber, R.; Do, J.; Roller, E.-M.; Zhang, T.; Schüller, V. J.; Nickels, P. C.; Feldmann, J.; Liedl, T. *Nat. Nanotechnol.* **2013**, *9*, 74–8.
- (20) Zhang, T.; Neumann, A.; Lindlau, J.; Wu, Y.; Pramanik, G.; Naydenov, B.; Jezek, F.; Schüder, F.; Huber, S.; Huber, M.; Stehr, F.; Högele, A.; Weil, T.; Liedl, T. *J. Am. Chem. Soc.* **2015**, *137*, 9776–9779.
- (21) Prinz, J.; Schreiber, B.; Olejko, L.; Oertel, J.; Rackwitz, J.; Keller, A.; Bald, I. *J. Phys. Chem. Lett.* **2013**, *4*, 4140–4145.
- (22) Kühler, P.; Roller, E.-M.; Schreiber, R.; Liedl, T.; Lohmüller, T.; Feldmann, J. *Nano Lett.* **2014**, *14*, 2914–9.
- (23) Thacker, V. V.; Herrmann, L. O.; Sigle, D. O.; Zhang, T.; Liedl, T.; Baumberg, J. J.; Keyser, U. F. *Nat. Commun.* **2014**, *5*, 3448.
- (24) Pilo-Pais, M.; Watson, A.; Demers, S.; LaBean, T. H.; Finkelstein, G. *Nano Lett.* **2014**, *14*, 2099–104.
- (25) Acuna, G. P.; Bucher, M.; Stein, I. H.; Steinhauer, C.; Kuzyk, A.; Holzmeister, P.; Schreiber, R.; Moroz, A.; Stefani, F. D.; Liedl, T.; Simmel, F. C.; Tinnefeld, P. *ACS Nano* **2012**, *6*, 3189–3195.
- (26) Acuna, G. P.; Möller, F. M.; Holzmeister, P.; Beater, S.; Lalkens, B.; Tinnefeld, P. *Science (Washington, DC, U. S.)* **2012**, *338*, 506–10.
- (27) Ko, S. H.; Du, K.; Liddle, J. A. *Angew. Chem., Int. Ed.* **2013**, *52*, 1193–1197.
- (28) Samanta, A.; Zhou, Y.; Zou, S.; Yan, H.; Liu, Y. *Nano Lett.* **2014**, *14*, 5052–7.
- (29) Kuzyk, A.; Schreiber, R.; Fan, Z.; Pardatscher, G.; Roller, E.-M.; Högele, A.; Simmel, F. C.; Govorov, A. O.; Liedl, T. *Nature* **2012**, *483*, 311–4.
- (30) Roller, E.-M.; Khorashad, L. K.; Fedoruk, M.; Schreiber, R.; Govorov, A. O.; Liedl, T. *Nano Lett.* **2015**, *15*, 1368–1373.
- (31) Gómez, D. E.; Giessen, H.; Davis, T. J. *J. Phys. Chem. C* **2014**, *118*, 23963–23969.
- (32) Savage, K. J.; Hawkeye, M. M.; Esteban, R.; Borisov, A. G.; Aizpurua, J.; Baumberg, J. J. *Nature* **2012**, *491*, 574–7.